

## DESCRIPTION

STEEL FOR STRUCTURAL PART OF AUTOMOBILE AND  
METHOD FOR PRODUCING THE SAME

## Technical Field

[0001] The present invention relates to steels suitably used for structural parts of automobiles and a method for producing the same. In particular, the present invention relates to improvements in formability, fatigue endurance after quenching, low temperature toughness, and resistance for hydrogen embrittlement of steels used as materials for forming and quenching suspension arms and axle beams. In the present invention, the term "steel" includes steel strips and steel tubes.

## Background Art

[0002] Patent Document 1 discloses a technique on a method for producing hot-rolled steels for body-reinforcing electric resistance welded tubes by induction hardening. The technique disclosed in Patent Document 1 can produce body-reinforcing electric resistance welded tubes required to have high strength, such as door guard beams and bumper reinforcements, which have high energy absorbability due to induction hardening. However, the technique has the problem of failing to achieve formability, fatigue endurance after quenching, and low temperature toughness required for suspension and chassis members.

[0003] Patent Document 2 discloses a technique on steel sheets for induction hardening and induction-hardened reinforcing members, which have excellent hardenability, hardened portions with toughness, and excellent high energy absorbability, and a method for producing these. The technique disclosed in Patent Document 2 can produce body-reinforcing members, such as center

pillars and bumper reinforcements, which have excellent high impact energy absorbability due to induction hardening. However, this technique has the problem of failing to achieve fatigue endurance after quenching and low temperature toughness which are required for suspension and chassis members.

[0004] Patent Document 3 discloses a technique on high toughness electric resistance welded tubes for heat treatment which can impart high strength and toughness by heat treatment. The technique disclosed in Patent Document 3 can produce automobile door reinforcements made from steel tubes by, for example, quenching after induction heating, the reinforcements having high strength and excellent low temperature toughness. However, this technique has the problem of failing to achieve fatigue endurance, resistance for hydrogen embrittlement, and corrosion fatigue endurance required for suspension and chassis members.

[0005] Patent Document 4 discloses a technique on low alloyed steel sheets having excellent properties of heat treatment by irradiation of a high energy density beam, high fatigue endurance after quenching, and high workability. The technique disclosed in Patent Document 4 can improve local fatigue endurance. However, this technique has the problem of failing to secure fatigue endurance required over the wholes of suspension and chassis members and to achieve resistance for hydrogen embrittlement and corrosion fatigue endurance required for these members.

[0006] Patent Document 5 discloses a technique on electric resistance welded tubes for hollow stabilizers which have excellent workability. The technique disclosed in Patent Document 5 can produce electric resistance welded tubes for hollow stabilizers having homogeneous metal structures in electric resistance welded portions and matrix portions and excellent workability due to induction heating and diameter reduction rolling of the electric resistance welded tubes.

However, this technique has the problem of failing to achieve fatigue endurance, resistance for hydrogen embrittlement, and corrosion fatigue endurance required for suspension and chassis members.

[0007] Patent Documents 6, 7, and 8 disclose techniques on high tension electric resistance welded tubes which have excellent resistance for hydrogen embrittlement. The techniques include increasing tensile strength of steel strips and then forming tubes to produce high tension steel tubes having excellent resistance for hydrogen embrittlement. However, these techniques have the problem of failing to achieve formability and fatigue endurance required for suspension and chassis members.

Patent Document 1: Japanese Examined Patent Application Publication No. 7-74382

Patent Document 2: Japanese Unexamined Patent Application Publication No. 2000-248338

Patent Document 3: Japanese Patent Publication No. 2605171

Patent Document 4: Japanese Unexamined Patent Application Publication No. 2000-248331

Patent Document 5: International Publication No. WO 02/070767 A1 pamphlet

Patent Document 6: Japanese Patent Publication No. 3111861

Patent Document 7: Japanese Patent Publication No. 3374659

Patent Document 8: Japanese Unexamined Patent Application Publication No. 2003-138316

## Disclosure of the Invention

**[0008]** An object of the present invention is to advantageously resolve the problems of the above-described conventional techniques and provide a steel having excellent formability, excellent fatigue endurance after quenching, excellent low temperature toughness, excellent resistance for hydrogen embrittlement, and excellent corrosion fatigue endurance which are required for suspension and chassis members, and a technique for producing the steel.

**[0009]** In the present invention, the term “excellent formability” means that a tensile test using a specimen of JIS No. 12 shows an elongation El of 20% or more. The term “excellent fatigue endurance after quenching” means that the maximum stress amplitude  $\sigma_f$  causing no fatigue failure in a completely reversed plane bending fatigue endurance test (stress ratio: -1.0) is 450 MPa or more. The term “excellent low temperature toughness” means that the fracture appearance transition temperature  $vTrs$  in a Charpy impact test (specimens: 1/4 size, 2 mm V-notch) is -40°C or less. The term “excellent resistance for hydrogen embrittlement” means that the failure time in a four-point bending test (load stress: 1180 MPa) in 0.1N hydrochloric acid is 200 hours or more. The term “excellent corrosion fatigue endurance” means that the fatigue life in a completely reversed plane bending fatigue test (stress ratio: -1) after a corrosion test is 1/2 or more of the number of cycles in a case without corrosion.

## Means for Solving the Problems

**[0010]** In order to obtain a steel being excellent in all the conflicting properties, such as formability, fatigue endurance after quenching, low temperature toughness, resistance for hydrogen embrittlement, and corrosion fatigue endurance, the inventors conducted systematic experimental study using various chemical compositions, microstructures before quenching, quenching methods and conditions, etc. As a result, it was found that when a chemical composition, strip

production conditions, material steel tube production conditions, and a structure before quenching are determined in specified limited ranges, a steel simultaneously satisfying all the formability, fatigue endurance after quenching, low temperature toughness, resistance for hydrogen embrittlement, and corrosion fatigue endurance, which are required for suspension and chassis members, can be obtained.

[0011] The present invention has been completed on the basis of this finding and additional study. Namely, the gist of the present invention is as follows:

(1) A steel for structural parts of automobiles having excellent formability, fatigue endurance after quenching, low temperature toughness, and resistance for hydrogen embrittlement, the steel having a composition containing, by mass, 0.18 to 0.29% of C, 0.06 to 0.45% of Si, 0.91 to 1.85% of Mn, 0.019% or less of P, 0.0029% or less of S, 0.015 to 0.075% of sol. Al, 0.0049% or less of N, 0.0049% or less of O, 0.0001 to 0.0029% of B, 0.001 to 0.019% of Nb, 0.001 to 0.029% of Ti, 0.001 to 0.195% of Cr, and 0.001 to 0.195% of Mo so that the carbon equivalent  $C_{eq}$  defined by the following equation (1):

$$C_{eq} = C + Mn/6 + Si/24 + Ni/40 + Cr/5 + Mo/4 + V/14 \quad (1)$$

(wherein C, Mn, Si, Ni, Cr, Mo, and V represent the contents (% by mass) of the respective elements) satisfies a value of 0.4 to less than 0.58, and the total  $\Sigma$  of multiplying factors including that for B according to Grossmann satisfies a value of 1.2 to less than 1.7, the balance being substantially composed of Fe, and the steel having a structure in which the ferrite grain diameter  $d_f$  corresponding to a circle is 1.1  $\mu m$  to less than 12  $\mu m$ , and the ferrite volume fraction  $V_f$  is 30% to less than 98%.

(2) The steel (1) for structural parts of automobiles further containing, by mass, at least one selected from 0.001 to 0.175% of Cu, 0.001 to 0.145% of Ni, and 0.001 to 0.029% of V in addition to the above composition:

(3) The steel (1) or (2) for structural parts of automobiles further containing 0.0001 to 0.0029% by mass of Ca in addition to the above composition.

(4) A method for producing a steel for structural parts of automobiles having excellent formability, fatigue endurance after quenching, low temperature toughness, and resistance for hydrogen embrittlement, the method including heating a steel slab at 1160°C to 1320°C, hot-finish-rolling the steel slab at a finisher delivery temperature of 750°C to 980°C, and then coiling the not-rolled steel at a coiling temperature of 560°C to 740°C after slow cooling for a time of 2 seconds or more to produce a hot-rolled steel strip, wherein the steel slab has a composition containing, by mass, 0.18 to 0.29% of C, 0.06 to 0.45% of Si, 0.91 to 1.85% of Mn, 0.019% or less of P, 0.0029% or less of S, 0.015 to 0.075% of sol. Al, 0.0049% or less of N, 0.0049% or less of O, 0.0001 to 0.0029% of B, 0.001 to 0.019% of Nb, 0.001 to 0.029% of Ti, 0.001 to 0.195% of Cr, and 0.001 to 0.195% of Mo so that the carbon equivalent  $C_{eq}$  defined by the following equation (1):

$$C_{eq} = C + Mn/6 + Si/24 + Ni/40 + Cr/5 + Mo/4 + V/14 \quad (1)$$

(wherein C, Mn, Si, Ni, Cr, Mo, and V represent the contents (% by mass) of the respective elements) satisfies a value of 0.4 to less than 0.58, and the total  $\Sigma$  of multiplying factors including that for B according to Grossmann satisfies a value of 1.2 to less than 1.7, the balance being preferably composed of Fe.

- (5) The method (4) for producing a steel for structural parts of automobiles, wherein the steel slab further contains, by mass, at least one selected from 0.001 to 0.175% of Cu, 0.001 to 0.145% of Ni, and 0.001 to 0.029% of V in addition to the above composition.
- (6) The method (4) or (5) for producing a steel for structural parts of automobiles, wherein the steel slab further contains 0.0001 to 0.0029% by mass of Ca in addition to the above composition.
- (7) A method for producing a steel tube for structural parts of automobiles having excellent formability, fatigue endurance after quenching, low temperature toughness, and resistance for hydrogen embrittlement, the method including making a tube from the hot-rolled steel strip by electric resistance welded tube making with a width reduction of hoop of 8% or less, the hot-rolled steel strip being produced by any one of the methods (4) to (6) and used as a raw material immediately after hot rolling or after pickling of the hot-rolled steel strip.

#### Brief Description of the Drawings

[0012] Fig. 1 is a graph showing the influences of carbon equivalent on the failure time in a four-point bending test in 0.1N hydrochloric acid after quenching, the fracture appearance transition temperature in a Charpy impact test, and the fatigue endurance in a completely reversed plane bending fatigue test.

[0013] Fig. 2 is a drawing schematically illustrating a stress loading method in a four-point bending test.

[0014] Fig. 3 is a graph showing the influence of the total  $\Sigma$  of multiplying factors for elements on the fatigue endurance  $\sigma_f$  in a completely reversed plane bending fatigue test.

**[0015]** Fig. 4 is a graph showing the influence of the ferrite grain diameter  $d_f$  corresponding to a circle of a structure before quenching on the fatigue endurance  $\sigma_f$  after quenching.

#### Best Mode for Carrying Out the Invention

**[0016]** The reasons for limiting the chemical component ranges of a steel of the present invention will be described. In the composition below, “% by mass” is simply shown by “%”.

C: 0.18 to 0.29%

**[0017]** C is an element necessary for securing fatigue endurance after quenching. However, at a C content of less than 0.18%, it is difficult to secure desired fatigue endurance, while at a C content exceeding 0.29%, resistance for hydrogen embrittlement degrades. Therefore, the C content is limited to a range of 0.18% to 0.29%. The C content is preferably 0.18% to 0.24%.

Si: 0.06% to 0.45%

**[0018]** Si is an element for accelerating ferrite transformation in a hot rolling step. In the present invention, in order to secure a ferrite volume fraction necessary for securing formability before quenching, the Si content must be 0.06% or more. At a Si content of less than 0.06%, the ferrite volume fraction becomes insufficient and formability degrades, while at a Si content exceeding 0.45%, the electric resistance weldability degrades, and the low-temperature toughness after quenching also degrades. Therefore, the Si content is limited to a range of 0.06% to 0.45%. The Si content is preferably 0.15% to 0.35%.

Mn: 0.91% to 1.85%

**[0019]** Mn is an element for suppressing ferrite transformation in a quenching step. In the present invention, in order to secure fatigue endurance after quenching which is required for a more than 90% tempered martensite structure, the Mn content must be 0.91% or more. At a Mn



content of less than 0.91%, a ferrite structure appears in a surface layer during quenching to fail to achieve desired fatigue endurance. On the other hand, at a Mn content exceeding 1.85%, the martensite transformation temperature ( $M_s$  point) of the steel decreases, and self-tempering (precipitation of micro carbide) of martensite in a quenching step is suppressed to increase the quenching strain of a quenched member and degrade resistance for hydrogen embrittlement. Therefore, the Mn content is limited to a range of 0.90% to 1.85%. The Mn content is preferably over 1.0% to 1.6%.

P: 0.019% or less

**[0020]** P is an element which segregates at austenite boundaries during quench and heating or segregates at boundaries between cementite and ferrite matrixes during cementite precipitation in a step of tempering martensite, thereby degrading low temperature toughness and resistance for hydrogen embrittlement. At a P content exceeding 0.019%, the adverse effect becomes significant. Therefore, the P content is limited to 0.019% or less. The P content is preferably 0.014% or less.

S: 0.0029% or less

**[0021]** S remains as an elongated MnS inclusion in steel to decrease formability, low temperature toughness, and fatigue endurance. In addition, S functions as an anode in corrosion under coating to accelerate local corrosion and entrance of hydrogen, thereby significantly decreasing resistance for hydrogen embrittlement and corrosion fatigue endurance. At a S content exceeding 0.0029%, the adverse effect becomes significant. Therefore, the upper limit of the S content is 0.0029% and preferably 0.0020%.

Sol. Al: 0.015% to 0.075%

**[0022]** Al is a deoxidizing element in steel making and an element for suppressing the growth of austenite grains in a hot-rolling step. In the present invention, in order to obtain a desired structure and grain diameter in combination with hot-rolling conditions, the sol. Al content must be 0.015% or more. At a sol. Al content of less than 0.015%, the above-described effect is not exhibited, while at a sol. Al content exceeding 0.075%, the effect is saturated, and an oxide inclusion is increased to degrade the production properties and fatigue endurance. Therefore, the sol. Al content is limited to 0.015% to 0.075%.

N: 0.0049% or less

**[0023]** N bonds to Ti and precipitates as TiN, but a variation thereof becomes a variation in surplus dissolved Ti, resulting in variations in strength and properties. Therefore, the content range must be strictly specified. At a N content exceeding 0.0049%, low temperature toughness degrades due to the precipitation of excessive TiN. Therefore, the upper limit of the N content is 0.0049%.

O: 0.0049% or less

**[0024]** O mainly remains as an inclusion in steel and decreases formability and fatigue endurance. At an O content exceeding 0.0049%, the adverse effect becomes significant. Therefore, the upper limit of the O content is 0.0049%. The O content is preferably 0.0020% or less.

B: 0.0001% to 0.0029%

**[0025]** B is an element necessary for securing hardenability without degrading resistance for hydrogen embrittlement. This effect is exhibited at a B content of 0.0001% or more. On the other hand, at a B content exceeding 0.0029%, the resistance for hydrogen embrittlement

degrades. Therefore, the B content is limited to 0.0001% to 0.0029%. The B content is preferably 0.0008% to 0.0018%.

Nb: 0.001% to 0.019%

**[0026]** Nb is an element which can make a fine structure in a hot-rolling step to form a desired structure and grain diameter due to a synergistic effect with AlN and which can suppress the growth of austenite grains during heating after forming to improve low temperature toughness after quenching. These effects are exhibited even at a low content of 0.001% or more, but formability before quenching degrades at a content exceeding 0.019%. Therefore, the Nb content is limited to a range of 0.001% to 0.019%.

Ti: 0.001% to 0.029%

**[0027]** Ti bonds to N and preferentially precipitates as TiN to effectively leave dissolved B, thereby contributing to the securement of hardenability. Furthermore, Ti decreases dissolved N to contribute to the securement of formability before quenching. These effects are exhibited at a content of 0.001% or more, but the formability before quenching and low temperature toughness degrade at a content exceeding 0.0029%. Therefore, the Ti content is limited to a range of 0.001% to 0.029%.

Cr: 0.001% to 0.195%

**[0028]** Cr is an element for improving hardenability and is contained for complementing the functions of Mn and B in the present invention. Also, a decrease in the Ms point due to the addition of Cr is lower than that by Mn, and thus Cr can suppress quenching strain. In addition, Cr hardly co-segregates with P at an austenite boundary during quenching and heating, and thus the addition of Cr has a small adverse effect on the resistance for hydrogen embrittlement. These

effects are expressed at a content of 0.001% or more, but formability before quenching degrades at a content exceeding 0.195%. Therefore, the Cr content is 0.001% to 0.195%.

Mo: 0.001% to 0.195%

**[0029]** Mo is an element for improving hardenability to complement the functions of Mn and B and decrease the potential of C in steel, thereby suppressing surface decarbonization during quenching and heating and significantly improving fatigue endurance after quenching. These effects are expressed at a content of 0.001% or more, but formability before quenching degrades at a content exceeding 0.195%. Therefore, the Mo content is 0.001% to 0.195%.

**[0030]** Besides the above-described basic composition, at least one selected from 0.001% to 0.175% of Cu, 0.001% to 0.145% of Ni, and 0.001% to 0.029% of V and/or 0.0001% to 0.0029% of Ca may be further contained.

**[0031]** Any one of Cu, Ni, and V is an element for improving resistance for hydrogen embrittlement and low temperature toughness, and at least one selected from these elements may be contained according to demand.

Cu: 0.001% to 0.175%

**[0032]** Cu is an element having the effect of suppressing the progress of corrosion by concentration as a metal element in a surface layer, particularly in a MnS anode portion, with the progress of corrosion, and the effect of suppressing the entrance of hydrogen into steel to improve resistance for hydrogen embrittlement. Cu may be added according to demand. At a content of 0.001% or more, these effects are expressed, but, at a content exceeding 0.175%, the problem of producing surface scars due to melted Cu during hot rolling increases. Therefore, the Cu content is preferably 0.001% to 0.175%.

Ni: 0.001% to 0.145%

[0033] Ni is an element having the effect of improving a strength-toughness balance and the effect of improving resistance for hydrogen embrittlement by concentration in a surface layer, and Ni may be contained according to demand. At a content of 0.001% or more, these effects are expressed, but, at a content exceeding 0.145%, austenite-ferrite transformation is suppressed during hot rolling, thereby failing to obtain a desired structure and decreasing formability before quenching. Therefore, the Ni content is preferably 0.001% to 0.145%.

V: 0.001% to 0.029%

[0034] V is an element having the function to complement the effect of Nb and may be contained according to demand. At a content of 0.001% or more, this effect is expressed, but, at a content exceeding 0.029%, formability before quenching degrades. Therefore, the V content is preferably 0.001% to 0.029%.

[0035] Besides the above-described composition, 0.0001% to 0.0029% of Ca may be further contained.

Ca: 0.0001% to 0.0029%

[0036] Ca is an element which precipitates as granular CaS in steel to decrease the amount of an elongated MnS inclusion, thereby improving formability, low temperature toughness, fatigue endurance, resistance for hydrogen embrittlement, and corrosion fatigue endurance, and Ca may be contained according to demand. At a content of 0.0001% or more, this effect is expressed, but, at a content exceeding 0.0029%, the adverse effect of a CaO inclusion on these properties becomes significant. Therefore, the Ca content is preferably 0.0001% to 0.0029%.

**[0037]** In the present invention, the above-described components are contained in the respective above-described ranges so that the carbon equivalent  $C_{eq}$  defined by the following equation (1):

$$C_{eq} = C + Mn/6 + Si/24 + Ni/40 + Cr/5 + Mo/4 + V/14 \quad (1)$$

(wherein C, Mn, Si, Ni, Cr, Mo, and V represent the contents (% by mass) of the respective elements) satisfies a value of 0.4 to less than 0.58, and the total  $\Sigma$  of multiplying factors including that for B according to Grossmann satisfies a value of 1.2 to less than 1.7.

Carbon equivalent  $C_{eq}$ : 0.4 to less than 0.58

**[0038]** When the carbon equivalent is less than 0.4, the desired hardening penetration and fatigue endurance cannot be obtained. On the other hand, when the carbon equivalent is 0.58 or more, resistance for hydrogen embrittlement after quenching and low temperature toughness degrade. Therefore, the carbon equivalent  $C_{eq}$  is 0.4 to less than 0.58, and preferably 0.44 to 0.54. A calculation according to the equation (1) is performed on the assumption that a value of an absent element is zero.

**[0039]** Fig. 1 shows the relations between the carbon equivalent  $C_{eq}$  and the four-point bending failure time in 0.1N hydrochloric acid after quenching, the fracture appearance transition temperature  $vTrs$  in a Charpy impact test, and completely reversed plane bending fatigue endurance  $\sigma_f$ . Fig. 1 indicates that when  $C_{eq}$  is 0.4 to less than 0.58, the steel is excellent in all the resistance for hydrogen embrittlement after quenching, the low temperature toughness, and the fatigue endurance.

**[0040]** A four-point bending test was carried out using a specimen of 5 mm in width and 80 mm in length which was cut out from a formed and quenched member. In the four-point bending test, the specimen was immersed in 0.1N hydrochloric acid and tested for a time of 200 hours at

a maximum with a load stress  $\sigma$  of 1180 MPa to determine the failure time and evaluate the resistance for hydrogen embrittlement. The load stress was calculated according to the following equation:

$$\sigma \text{ (MPa)} = (12E t \delta)/(3H^2 - 4A^2)$$

(wherein E represents the modulus of longitudinal elasticity ( $2.06 \times 10^5$  MPa); t, the thickness (mm) of the specimen; H, the distance (mm) between both end fulcrums;  $A = (H - h)/2$  wherein h is the distance (mm) between intermediate fulcrums; and  $\delta$ , a displacement (mm) at the center of the specimen). Fig. 2 shows a stress loading method in the four-point bending test.

[0041] Also, a 2 mm V-notch impact specimen of a 1/4 size (thickness: 2.5 mm) was obtained from the formed and quenched member and measured by a Charpy impact test to determine the fracture appearance transition temperature vTrs and evaluate the low temperature toughness.

[0042] Furthermore, a specimen of 30 mm in width and 90 mm in length was cut out from the formed and quenched member and measured by a completely reversed plane bending fatigue test to evaluate fatigue endurance. In the completely reversed plane bending fatigue test, the maximum stress amplitude  $\sigma_f$  causing no fatigue failure was determined as fatigue endurance with a stress ratio of -1.0 under completely reversed plane bending, a repetition rate of 25 Hz, and a maximum number of cycles of  $10^7$  cycles.

Total x of multiplying factors including that for B according to Grossmann: 1.2 to less than 1.7

[0043] The multiplying factors according to Grossmann are material parameters required to be controlled for achieving the desired formability of an original sheet, hardness after quenching, and fatigue endurance. In the present invention, as the multiplying factors according to Grossmann, for example, the values for the respective elements described in Table 3 of Iron & Steel

Material Science, Leslie, Maruzen, pp. 402-405 are used. In other words, the value of each element is determined according to the content thereof, and the total  $x$  of the elements is determined. As the multiplying factor for C among the multiplying factors according to Grossmann, the value for ASTM grain No. 7 is used. When dissolved B is contained in an amount larger than the N equivalent with consideration of fixing by TiN, the multiplying factor for B which has not been specified is considered as 0.2 regardless of the content.

**[0044]** When the total  $x$  is less than 1.2, hardness after quenching decreases, and it is thus impossible to obtain the excellent fatigue endurance after quenching, e.g., a maximum stress amplitude  $\sigma_f$  of 450 MPa or more without fatigue failure in the completely reversed plane bending fatigue test. On the other hand, when the total  $x$  is 1.7 or more, the ferrite volume fraction of steel is less than 30%, the formability of an original sheet decreases, and a portion where the thickness is locally decreased becomes a stress concentration portion, thereby failing to achieve excellent fatigue endurance, e.g., an  $\sigma_f$  of 450 MPa or more after quenching. Therefore, the total  $x$  of multiplying factors including that for B according to Grossmann is limited to 1.2 to less than 1.7. Fig. 3 shows the relation between the total  $x$  of multiplying factors for the elements and the fatigue endurance  $\sigma_f$  in the completely reversed plane bending fatigue test.

**[0045]** The balance other than the above-described components is substantially composed of Fe.

**[0046]** The steel of the present invention has a structure in which the ferrite grain diameter  $d_f$  corresponding to a circle is 1.1  $\mu\text{m}$  to less than 12  $\mu\text{m}$ , and the ferrite volume fraction  $V_f$  is 30% to less than 98%.



Ferrite grain diameter  $d_f$  corresponding to a circle: 1.1  $\mu\text{m}$  to less than 12  $\mu\text{m}$

[0047] The microstructure of a raw material (steel) before quenching is an important material parameter for securing excellent formability, high fatigue endurance after quenching, and the like. When the ferrite grain diameter  $d_f$  corresponding to a circuit is less than 1.1  $\mu\text{m}$ , desired formability cannot be secured, and a portion where the thickness is locally decreased becomes a stress concentration portion, thereby greatly decreasing fatigue endurance after quenching. On the other hand, when the  $d_f$  is 12  $\mu\text{m}$  or more, particularly the hardenability of a raw material surface decreases, and thus fatigue endurance greatly decreases. Therefore, the ferrite grain diameter  $d_f$  corresponding to a circle of the steel is limited to 1.1  $\mu\text{m}$  to less than 12  $\mu\text{m}$ .

[0048] Fig. 4 shows the relation between the ferrite grain diameter  $d_f$  corresponding to a circuit of a raw material (steel) structure before quenching and the fatigue endurance  $\sigma_f$  in the completely reversed plane bending fatigue test after quenching. The figure indicates that when the ferrite grain diameter  $d_f$  corresponding to a circuit of a raw material (steel) structure before quenching is 1.1  $\mu\text{m}$  to less than 12  $\mu\text{m}$ , high fatigue endurance, e.g., an  $\sigma_f$  of 450 MPa or more, is obtained. In particular, when  $d_f$  is in a range of 2.0 to 7.9  $\mu\text{m}$ , a high fatigue endurance of 500 MPa or more is obtained. The term “ferrite” includes Polygonal ferrite, Acicular ferrite, Widmannstaetten ferrite, and Bainitic ferrite.

[0049] In the present invention, the ferrite grain diameter  $d_f$  corresponding to a circle is determined by a method in which the area of each ferrite grain is measured by processing an image of the structure, the area of each ferrite grain is converted to a diameter corresponding to a circle, and the obtained ferrite grain diameters corresponding to a circle are averaged. Like the steel used in the present invention, a material having hardenability contains ferrite grains having

no regular form. Therefore, the used ferrite grain diameter is the diameter corresponding to a circle determined by image processing, not the diameter determined by a cutting method.

**[0050]** A second phase other than the ferrite phase preferably includes carbide, pearlite, bainite, martensite, or a mixture thereof. Like in the ferrite, in the second phase, the average grain diameter  $d_s$  corresponding to a circle is preferably 1.1  $\mu\text{m}$  to less than 12  $\mu\text{m}$ .

Ferrite volume fraction  $V_f$ : 30% to less than 98%

**[0051]** When the volume fraction  $V_f$  of the raw material (steel) before quenching is less than 30%, desired formability cannot be secured, and a portion where the thickness is locally decreased becomes a stress concentration portion, thereby greatly decreasing fatigue endurance after quenching. On the other hand, within the above-described component ranges, it is difficult to secure a ferrite volume fraction of 98% or more. Therefore, this value is the upper limit. The ferrite volume fraction is determined by a method in which a section is etched with nital and then observed in at least two fields of view and imaged by a scanning electron microscope (SEM) with a magnification of  $\times 1000$ , grain boundaries and the second phase are separated, and the ferrite volume fraction is determined by image processing.

**[0052]** Next, a method for producing a hot-rolled steel strip of the present invention will be described.

**[0053]** First, preferably, molten steel having the above-described composition is refined by a usual refining method such as a converter or the like and then formed in a steel slab by a usual casting method such as continuous casting or the like. Next, the steel slab having the above composition is heated and hot-rolled to form a hot-rolled steel strip.

**[0054]** Next, the preferred hot rolling conditions for the steel slab will be described.

Slab heating temperature: 1160°C to 1320°C

[0055] In the present invention, elements which form slightly soluble carbonitrides, such as Nb, Ti, and the like, are used as essential elements. Therefore, when the slab heating temperature is lower than 1160°C, re-dissolution of carbonitrides becomes locally insufficient, and the ferrite grain diameter after hot rolling partially exceeds 12  $\mu\text{m}$ , thereby decreasing workability before quenching. On the other hand, when the slab heating temperature exceeds 1320°C, the surface quality of final products, such as steel tubes and steel strips, degrades. Therefore, the slab heating temperature is preferably 1160°C to 1320°C and more preferably 1180°C to 1280°C.

Finisher delivery temperature: 750°C to 980°C

[0056] The finisher delivery temperature of hot rolling is an important production parameter which determines the ferrite grain diameter after hot rolling. When the finisher delivery temperature is lower than 750°C, rolling in the ferrite region is caused, and rolling strain remains after coiling to decrease formability before quenching. On the other hand, when the finisher delivery temperature exceeds 980°C, the ferrite grains coarsen, thereby decreasing formability before quenching. Therefore, the finisher delivery temperature is preferably 750°C to 980°C and more preferably 820°C to 940°C.

Slow cooling time before coiling: 2 seconds or more

[0057] After finish rolling of hot rolling is finished, the steel strip is not immediately coil, but a slow cooling time of 2 seconds or more is preferably secured before coiling. The slow cooling after the end of finish rolling means cooling at a cooling rate of 20°C/s or less. Consequently, ferrite transformation can be sufficiently performed to improve formability before quenching.

Coiling temperature: 560°C to 740°C

[0058] The coiling temperature after the finish of hot rolling is an important production parameter which determines the ferrite volume fraction after hot rolling. When the coiling temperature is lower than 560°C, the desired ferrite volume fraction cannot be obtained, and thus formability before quenching decreases. The higher the coiling temperature within the specified range, the more the formability before quenching is improved. However, when the coiling temperature exceeds 740°C, the C amount in a surface layer significantly decreases, and thus fatigue endurance after quenching degrades. Therefore, the coiling temperature is preferably 560°C to 740°C and more preferably 600°C to 700°C.

[0059] Even with a steel slab in which it is difficult to produce a desired ferrite structure because of high hardenability, therefore, the above-described production method produces a hot-rolled steel strip having a desired microstructure most suitable for formability before quenching and fatigue endurance after quenching, e.g., an average ferrite grain diameter corresponding to a circle of 1.1  $\mu\text{m}$  to 12  $\mu\text{m}$  and a ferrite volume fraction of 30% to 98%. As a result, it is possible to produce a hot-rolled steel strip for structural parts of automobiles having excellent formability, fatigue endurance after quenching, low temperature toughness, and resistance for hydrogen embrittlement.

[0060] In the present invention, by using the hot-rolled steel strip produced by the above-described method, a steel tube for structural parts of automobiles having excellent formability, fatigue endurance after quenching, low temperature toughness, and resistance for hydrogen embrittlement can be produced by electric resistance welded tube making under proper conditions.

[0061] Next, preferred conditions for producing steel tubes will be described.

**[0062]** In the present invention, the hot-rolled steel strip produced under the above-described conditions is used as a raw material. The raw material may be used immediately after hot rolling or after descaling by pickling. The raw material immediately after hot rolling or after pickling is preferably formed in a steel tube by electric resistance welded tube making with a width reduction of hoop of 8% or less.

Conditions for electric resistance welded tube making: width reduction of hoop of 8% or less

**[0063]** When the steel strip is formed in a steel tube by continuous roll forming and electric resistance welded tube making, the width reduction of hoop is an important production parameter for securing desired formability before quenching. When the width reduction of hoop exceeds 8%, formability significantly decreases with tube making, and thus necessary formability before quenching cannot be obtained. Therefore, the width reduction of hoop is preferably 8% or less (including 0%). The width reduction of hoop is defined by the following equation:

Width reduction of hoop (%) =  $\frac{[(\text{width of untreated steel strip}) - \pi \{(\text{outer diameter of product} - \text{wall thickness of product})\}]}{[\pi \{(\text{outer diameter of product}) - (\text{wall thickness of product})\}]} \times 100$

**[0064]** In the method for producing the steel tube of the present invention, the raw material is not limited to the hot-rolled steel strip. Of course, a cold-rolled annealed steel strip produced by cold-rolling and annealing the hot-rolled steel strip produced by the above-described method for producing the hot-rolled steel strip, or a surface-treated steel strip further subjected to various surface treatments may be used. Instead of the electric resistance welded tube making, roll forming, press-section closing of a cut plate, or a tube making process combined with cold-, warm-, or hot-reduction, heat treatment, and the like may be used. Furthermore, instead of electric resistance welding, laser welding, arc welding, plasma welding, or the like may be used.

## EXAMPLES

### EXAMPLE 1

[0065] Each of the 26 types of steel slabs A to Z shown in Table 1 was re-heated to the slab heating temperature shown in Table 3 and then hot-rolled under the conditions shown in Table 3 to form a hot-rolled steel strip of 2.6 mm in thickness. The resulting hot-rolled steel strip was pickled, slitted, and then subjected to electric resistance welded tube making by roll forming and electric resistance welding to form a welded steel tube having an outer diameter of 101.6 mm. The width reduction of hoop in tube making was as shown in Table 3. Table 2 shows the multiplying factors for elements in each steel and the total thereof.

[0066] Furthermore, a specimen for structure observation was obtained from each hot-rolled steel strip, polished, corroded, observed with SEM ( $\times 1000$ ), and imaged to measure the ferrite volume fraction, the average ferrite grain diameter corresponding to a circle, and the second phase grain diameter corresponding to a circle by image processing. The grain diameter corresponding to a circle was determined by a method in which the area of each grain was measured, the diameter of a circle corresponding to the area was determined as the diameter corresponding to a circle, and the diameters of the grains were averaged.

[0067] The resulting steel tube was formed in an axle beam with a closed section and then quenched by heating at about 920°C in a continuous furnace with a controlled atmosphere and then water cooling. After the quenching, a cross-section hardness test, a completely reversed plane bending fatigue test, a Charpy impact test, a four-point bending test, and a completely reversed plane bending fatigue test after a corrosion test were carried out. The test methods were as follows:

(1) Cross-section hardness test

A specimen was cut out from a formed and quenched member, and the Vickers hardness (load: 10 kgf) was measured over the entire region in the thickness direction. The measured values were averaged to determine the cross-section hardness of the member after quenching.

(2) Completely reversed plane bending fatigue test

In the completely reversed plane bending fatigue test, a specimen of 30 mm in width and 90 mm in length was cut out from a formed and quenched member and tested at a stress ratio of -1.0 under completely reversed plane bending, a repetition rate of 25 Hz, and a maximum number of cycles of  $10^7$  cycles to determine the maximum stress amplitude  $\sigma_f$  as fatigue endurance and evaluate fatigue endurance after quenching.

(3) Charpy impact test

In the Charpy impact test, a 2 mm V-notch impact specimen of 1/4 size (thickness: 2.5 mm) was obtained from a formed and quenched member and tested to determine the fracture appearance transition temperature  $vTrs$  and evaluate low temperature toughness.

(4) Four-point bending test

In the four-point bending test, a specimen of 5 mm in width and 80 mm in length was cut out from a formed and quenched member and immersed in 0.1N hydrochloric acid to determine the failure time by the test with a load stress  $\sigma$  of 1180 MPa for a time of 200 hours at a maximum as shown in Fig. 2, for evaluating resistance for hydrogen embrittlement. The load stress was calculated according to the following equation:

$$\sigma \text{ (MPa)} = (12E t \delta) / (3H^2 - 4A^2)$$

(wherein  $E$  represents the modulus of longitudinal elasticity ( $2.06 \times 10^5$  MPa);  $t$ , the thickness (mm) of the specimen;  $H$ , the distance (mm) between both end fulcrums;  $A = (H - h)/2$  wherein  $h$  is the distance (mm) between intermediate fulcrums; and  $\delta$ , an amount of displacement (mm) at the center of the specimen).

(5) Completely reversed plane bending fatigue test after corrosion test

In the completely reversed plane bending fatigue test after a corrosion test, a specimen of 30 mm in width and 90 mm in length was cut out from a formed and quenched member and subjected to a salt spray test for 20 days according to the prescriptions of JIS Z2371. Then, the completely reversed plane bending fatigue test (stress ratio: -1, repetition rate: 25 Hz) was carried out with stress amplitude  $\sigma_f$  to determine the fatigue life. When the fatigue life was 1/2 or more of the number of cycles of an uncorroded specimen, it was evaluated as O, and other cases were evaluated as X.

[0068] Each of the steel tubes before quenching was subjected to a tensile test of the untreated tube using a tensile specimen of JIS No. 12 to determine elongation El and evaluate formability. The results are shown in Tale 4.

[0069] In all examples of the present invention, the elongation El of the untreated tube is 20% or more (JIS No. 12 specimen), the cross-section hardness HV(10) after quenching is 350 to 550, the completely reversed plane bending fatigue endurance  $\sigma_f$  is 500 MPa or more, the fracture appearance transition temperature  $vTrs$  in the Charpy impact test is  $-40^\circ\text{C}$  or less, the four-point bending failure time in 0.1N hydrochloric acid is 200 hours or more, and the fatigue life in the corrosion fatigue test is 1/2 or more of the number of cycles of the uncorroded specimen. Therefore, excellent formability, fatigue endurance, low temperature toughness, resistance for hydrogen embrittlement, and corrosion fatigue endurance are exhibited.



**[0070]** On the other hand, in Comparative Example Nos. 5 to 26 in which any one of the steel composition, the carbon equivalent  $C_{eq}$ , and the total  $x$  of multiplying factors is out of the range of the present invention, any one of the formability, fatigue endurance, low temperature toughness, resistance for hydrogen embrittlement, and corrosion fatigue endurance decreases. In Comparative Example Nos. 5, 9, and 14 in which the contents of C, Mn, and B, respectively, in the steel composition are lower than the ranges of the present invention, the cross-section hardness Hv after quenching is less than 350, and  $\sigma_f$  after quenching is as low as less than 450 MPa. In Comparative Example Nos. 6, 10, and 15 in which the contents of C, Mn, and B, respectively, in the steel composition exceed the ranges of the present invention, the four-point bending failure time in 0.1N hydrochloric acid is less than 200 hours, and thus the resistance for hydrogen embrittlement degrades. In Comparative Example No. 7 in which the Si content is lower than the range of the present invention, the ferrite volume fraction is as low as less than 30%, and the elongation El of the untreated tube is as low as less than 20%. On the other hand, in Comparative Example No. 8 in which the Si content exceeds the range of the present invention,  $v_{Trs}$  is -40°C or higher, and thus low temperature toughness degrades. In Comparative Example Nos. 11, 12, and 13 in which the contents of P, S, and O, respectively, exceed the ranges of the present invention, resistance for hydrogen embrittlement, fatigue endurance, or untreated tube El is low.

**[0071]** In Comparative Example Nos. 16, 17, 18, 19, 20, and 21 in which the contents of Nb, Ti, Cr, Mo, Ni, and V, respectively, exceed the ranges of the present invention, the untreated tube El is as low as less than 20%, and thus formability degrades. In Comparative Example No. 22 in which the Ca content exceeds the range of the present invention, El, fatigue endurance, low temperature toughness, and resistance for hydrogen embrittlement are low. In Comparative

Example Nos. 23 and 25 in which the carbon equivalent  $C_{eq}$  or the total  $x$  of multiplying factors exceeds the range of the present invention, the untreated tube EI is low, the hardness HV(10) after quenching is as high as over 550,  $vTrs$  is high, and resistance for hydrogen embrittlement decreases. In Comparative Example Nos. 24 and 26 in which  $C_{eq}$  or  $x$  is lower than the range of the present invention, the ferrite grain diameter corresponding to a circle is as large as 12  $\mu m$  or more, the hardness HV(10) after quenching is as low as less than 350, and the completely reversed plane bending fatigue endurance  $\sigma_f$  after quenching is as low as less than 450 MPa.

## EXAMPLE 2

[0072] A hot-rolled steel strip was produced using a steel slab having each of the compositions of steels A to D shown in Table 1 under the hot rolling conditions shown in Table 5. The hot-rolled steel strip was descaled by pickling and then formed in an electric resistance welded tube under the tube making conditions shown in Table 5. Some of the hot-rolled steel strips were used immediately after hot rolling (with scale). Also, some of the hot-rolled steel strips were formed in tubes after cold rolling and annealing, and Zn or Al surface treatment. For some of the hot-rolled steel strips, press welding or roll forming-welding was used instead of electric resistance welded tube making. Also, some of the steel tubes were plated with Zn after tube making. Furthermore, some of the steel tubes were subjected to warm- or hot-reduction.

[0073] Each of the steel tubes used as tubular materials was formed in an axle beam (or formed in a suspension arm in No. 41) and then heated to about 920°C in a continuous furnace with a controlled atmosphere, followed by water cooling. Some of the members were subjected to shot blasting or shot peening after quenching.

[0074] For the resulting members after quenching, the cross-section hardness test, the completely reversed plane bending test, the Charpy impact test, the four-point bending test in 0.1N

hydrochloric acid, and the completely reversed plane bending fatigue test after the salt water spray test for 20 days (JIS Z2371) were conducted. The test methods were the same as in EXAMPLE 1. The results are shown in Table 6.

[0075] In all examples of the present invention, the elongation El of the untreated tube is 20% or more (JIS No. 12 specimen), the cross-section hardness HV(10) after quenching is 350 to 550, the completely reversed plane bending fatigue endurance  $\sigma_f$  is 500 MPa or more, the fracture appearance transition temperature  $vTrs$  in the Charpy impact test is  $-40^{\circ}\text{C}$  or less, the four-point bending failure time in 0.1N hydrochloric acid is 200 hours or more, and the decreased fatigue life in the corrosion fatigue test is 1/2 or more of the number of cycles of an uncorroded specimen. Therefore, excellent formability, fatigue endurance, low temperature toughness, resistance for hydrogen embrittlement, and corrosion fatigue endurance are exhibited.

[0076] In Example No. 28 in which the slab heating temperature is lower than the preferred range of the present invention, Example No. 30 in which the finisher delivery temperature is higher than the preferred range of the present invention, Example No. 31 in which the finisher deliver temperature is lower than the preferred range of the present invention, Example No. 33 in which the slow cooling time on a run out table of hot rolling is shorter than the preferred range of the present invention, and Example No. 35 in which the coiling temperature is lower than the preferred range of the present invention, the ferrite grain diameter  $d_f$  is larger than  $12\text{ }\mu\text{m}$  or the ferrite volume fraction  $V_f$  is less than 30%, the elongation of the untreated tube is as low as less than 20% to decrease formability, and the completely reversed plane bending fatigue endurance  $\sigma_f$  after quenching is as low as less than 450 MPa except in Example No. 33. In Example No. 36 in which the hot-rolled strip coiling temperature is higher than the preferred range of the present invention, the elongation El of the untreated tube is as high as 20% or more, but  $\sigma_f$  is low due to

surface decarbonization. In Example No. 38 in which the width reduction of hoop in tube making is higher than the preferred range of the present invention and the coiling temperature is lower than the preferred range of the present invention, the elongation El of the untreated tube is as low as 15%, and  $\sigma_f$  is also low.

[0077] In Example No. 40, pickling after hot rolling was omitted; in Example No. 41, the suspension arm of  $\varnothing 60.5 \times 2.6$  t was formed and then quenched; in Example No. 42, the steel strip subjected to cold rolling and annealing after hot rolling was formed in an electric resistance welded tube; in Example No. 43, the steel strip was press-formed and then welded (arc, laser, or plasma) to a closed section; in Example No. 44, the steel strip was roll-formed to a closed section and then welded; in Example Nos. 45 and 46, the black plate subjected to Zn or Al plating after quenching was formed in an electric resistance welded tube; in Example No. 47, the electric resistance welded tube was plated with Zn and then quenched; in Examples Nos. 48 and 49, the electric resistance welded tube was subjected to hot- or warm-reduction; in Example No. 50, the untreated tube was heated and quenched during forming; in Example No. 51, shot blasting was performed after quenching; and in Example No. 52, shot peening was performed after quenching. In all Example Nos. 40 to 52 which are examples of the present invention, the elongation El of the untreated tube is 20% or more (JIS No. 12 specimen), the cross-section hardness HV(10) after quenching is 350 to 550, the completely reversed plane bending fatigue endurance  $\sigma_f$  is 500 MPa or more, the fracture appearance transition temperature  $vTrs$  in the Charpy impact test is -40°C or less, the four-point bending failure time in 0.1N hydrochloric acid is 200 hours or more, and a decrease in the fatigue life in the corrosion fatigue test is less than 1/2 of the number of cycles of an uncorroded specimen. Therefore, excellent formability, fatigue endurance, low tem-

perature toughness, resistance for hydrogen embrittlement, and corrosion fatigue endurance are exhibited.

#### Industrial Applicability

[0078] According to the present invention, it is possible to easily produce a steel for structural parts of automobiles at low cost, the steel having excellent formability, excellent fatigue endurance after quenching, excellent low temperature toughness, excellent resistance for hydrogen embrittlement, and excellent corrosion fatigue endurance which are required for suspension and chassis members.

Table 1

Steel	Chemical components (% by mass)																	Carbon equivalent Ceq
	C	Si	Mn	P	S	Sol.Al	N	O	B	Nb	Ti	Cr	Mo	Cu	Ni	V	Ca	
A	0.21	0.15	1.44	0.011	0.0024	0.035	0.0032	0.0017	0.0001	0.006	0.023	0.137	0.168	0.030	0.034	0.015	0.0003	0.528
B	0.23	0.18	1.28	0.014	0.0021	0.043	0.0017	0.0016	0.0015	0.002	0.004	0.081	0.022	0.142	-	-	0.0022	0.473
C	0.18	0.41	1.80	0.013	0.0013	0.055	0.0042	0.0008	0.0013	0.014	0.025	0.032	0.132	-	-	-	-	0.536
D	0.20	0.20	1.31	0.010	0.0009	0.030	0.0035	0.0012	0.0011	0.015	0.013	0.149	0.102	-	-	-	-	0.482
E	0.16	0.22	1.26	0.008	0.0005	0.021	0.0024	0.0012	0.0008	0.016	0.018	0.186	0.078	0.084	-	-	0.0006	0.436
F	0.32	0.08	0.92	0.002	0.0023	0.016	0.0037	0.0023	0.0013	0.011	0.022	0.012	0.074	-	-	-	-	0.498
G	0.19	0.03	1.52	0.004	0.0016	0.072	0.0015	0.0016	0.0009	0.008	0.015	0.008	0.032	-	-	0.012	-	0.454
H	0.22	1.23	1.13	0.016	0.0017	0.037	0.0037	0.0011	0.0016	0.009	0.013	0.168	0.108	-	-	-	-	0.520
I	0.27	0.32	0.84	0.014	0.0011	0.053	0.0046	0.0009	0.0014	0.001	0.013	0.132	0.163	-	-	-	-	0.490
J	0.22	0.25	1.94	0.018	0.0019	0.067	0.0042	0.0013	0.0012	0.022	0.012	0.154	0.167	-	0.055	-	-	0.627
K	0.21	0.17	1.34	0.023	0.0009	0.035	0.0031	0.0015	0.0019	0.009	0.018	0.067	0.122	0.120	-	-	-	0.484
L	0.19	0.18	1.20	0.015	0.0045	0.022	0.0029	0.0013	0.0016	0.011	0.007	0.069	0.156	-	-	-	0.0011	0.450
M	0.21	0.26	1.08	0.014	0.0011	0.036	0.0026	0.0056	0.0023	0.016	0.004	0.089	0.132	-	-	-	-	0.452
N	0.18	0.14	1.16	0.017	0.0078	0.029	0.0040	0.0070	0.0000	0.015	0.013	0.182	0.176	0.078	-	-	-	0.460
O	0.25	0.35	0.97	0.014	0.0006	0.074	0.0019	0.0015	0.0032	0.019	0.015	0.122	0.155	-	-	0.023	-	0.491
P	0.26	0.26	1.36	0.013	0.0003	0.051	0.0033	0.0011	0.0015	0.032	0.005	0.156	0.186	-	-	-	-	0.575
Q	0.21	0.40	1.55	0.015	0.0012	0.020	0.0040	0.0021	0.0018	0.027	0.044	0.081	0.176	-	-	-	0.0014	0.545
R	0.22	0.24	1.16	0.013	0.0008	0.057	0.0046	0.0006	0.0011	0.016	0.022	0.254	0.143	-	0.176	-	-	0.514
S	0.26	0.09	1.34	0.011	0.0014	0.044	0.0047	0.0011	0.0010	0.015	0.029	0.116	0.275	-	-	-	-	0.579
T	0.20	0.10	1.24	0.012	0.0021	0.025	0.0019	0.0015	0.0009	0.006	0.014	0.181	0.122	-	0.167	-	-	0.482
U	0.20	0.24	1.37	0.011	0.0009	0.067	0.0037	0.0015	0.0011	0.015	0.002	0.116	0.182	-	-	0.035	-	0.510
V	0.23	0.37	1.33	0.008	0.0018	0.035	0.0032	0.0013	0.0024	0.014	0.021	0.156	0.146	-	-	-	0.0045	0.535
W	0.23	0.40	1.60	0.018	0.0026	0.045	0.0023	0.0016	0.0017	0.012	0.011	0.120	0.160	-	-	0.020	-	0.579
X	0.18	0.07	1.25	0.012	0.0015	0.018	0.0017	0.0015	0.0005	0.011	0.021	0.040	0.016	-	-	-	-	0.403
Y	0.28	0.40	1.78	0.013	0.0008	0.033	0.0016	0.0015	0.0012	0.008	0.014	0.002	0.122	-	0.156	-	-	0.628
Z	0.19	0.12	1.10	0.010	0.0010	0.038	0.0032	0.0012	0.0011	0.013	0.008	0.040	0.023	-	-	0.020	-	0.393

Table 2

Steel	Multiplying factor for each element																	Total of multi- plying factors $\chi$
	C	Si	Mn	P	S	Sol. Al	N	O	B	Nb	Ti	Cr	Mo	Cu	Ni	V	Ca	
A	0.188	0.043	0.794	0.011	-	0.017	-	-	0.200	-	-0.018	0.108	0.170	-	0.005	0.061	-	1.579
B	0.208	0.052	0.733	0.011	-	0.022	-	-	0.200	-	-	0.069	0.025	-	-	-	-	1.320
C	0.158	0.110	0.908	0.011	-	0.028	-	-	0.200	-	-0.018	0.027	0.143	-	-	-	-	1.567
D	0.179	0.057	0.741	0.011	-	0.017	-	-	0.200	-	-0.008	0.115	0.114	-	-	-	-	1.426
E	0.133	0.062	0.725	-	-	0.012	-	-	0.200	-	-0.008	0.143	0.083	-	-	-	-	1.350
F	0.281	0.024	0.609	-	-	0.006	-	-	0.200	-	-0.018	0.009	0.083	-	-	-	-	1.194
G	0.169	0.009	0.822	-	-	0.039	-	-	0.200	-	-0.008	-	0.037	-	-	0.061	-	1.329
H	0.198	0.268	0.675	0.011	-	-	-	-	0.200	-	-0.008	0.129	0.114	-	-	-	-	1.604
I	0.241	0.088	0.580	0.011	-	0.028	-	-	0.200	-	-0.008	0.108	0.170	-	-	-	-	1.418
J	0.198	0.070	0.946	0.011	-	0.033	-	-	0.200	-	-0.008	0.122	0.170	-	0.008	-	-	<u>1.750</u>
K	0.188	0.049	0.757	0.022	-	0.017	-	-	0.200	-	-0.008	0.053	0.134	-	-	-	-	1.412
L	0.169	0.052	0.702	0.011	-	0.012	-	-	0.200	-	-	0.053	0.161	-	-	-	-	1.360
M	0.188	0.073	0.662	0.011	-	0.017	-	-	0.200	-	-	0.069	0.143	-	-	-	-	1.363
N	0.158	0.041	0.687	0.011	-	0.012	-	-	-	-	-0.008	0.143	0.179	-	-	-	-	1.223
O	0.225	0.095	0.627	0.011	-	0.039	-	-	0.200	-	-0.008	0.101	0.161	-	-	0.097	-	1.548
P	0.233	0.073	0.765	0.011	-	0.028	-	-	0.200	-	-	0.122	0.188	-	-	-	-	1.620
Q	0.188	0.107	0.828	0.011	-	0.012	-	-	0.200	-	-0.034	0.069	0.179	-	-	-	-	1.560
R	0.198	0.067	0.687	0.011	-	0.028	-	-	0.200	-	-0.018	0.187	0.152	-	0.026	-	-	1.538
S	0.233	0.027	0.757	0.011	-	0.022	-	-	0.200	-	-0.018	0.093	0.258	-	-	-	-	1.583
T	0.179	0.029	0.718	0.011	-	0.012	-	-	0.200	-	-0.008	0.143	0.134	-	0.024	-	-	1.442
U	0.179	0.067	0.765	0.011	-	0.017	-	-	0.200	-	-	0.093	0.188	-	-	0.137	-	1.657
V	0.208	0.100	0.749	-	-	0.017	-	-	0.200	-	-0.018	0.122	0.152	-	-	-	-	1.530
W	0.208	0.107	0.846	0.011	-	0.022	-	-	0.200	-	-0.008	0.101	0.170	-	-	0.097	-	<u>1.754</u>
X	0.158	0.021	0.718	0.011	-	0.006	-	-	0.200	-	-0.018	0.036	0.013	-	-	-	-	<u>1.145</u>
Y	0.250	0.107	0.902	0.011	-	0.017	-	-	0.200	-	-0.008	-	0.134	-	0.023	-	-	1.636
Z	0.169	0.035	0.669	0.011	-	0.017	-	-	0.200	-	-	0.036	0.025	-	-	0.097	-	1.259

Table 3

No.	Steel	Hot rolling conditions				Tube making
		Slab heating temperature (°C)	Finisher temperature (°C)	Slow cooling time (s)	Coiling temperature (°C)	Width reduction of hoop (%)
1 - 25	A - Z	About 1240	About 910	About 4	About 660	About 3



Table 4

No.	Steel	Ferrite gain diameter corresponding to circle, $d_f$ ( $\mu\text{m}$ )	Second phase diameter corresponding to circle, $d_s$ ( $\mu\text{m}$ )	Ferrite volume fraction, $V_f$ (%)	Elongation of untreated tube El (%)	Cross-section hardness after quenching HV(10)	Completely reversed plane bending fatigue endurance, $\sigma_f$ (MPa)	Fracture appearance transition temperature in Charpy impact test $v_{Trs}$ ( $^{\circ}\text{C}$ )	Four-point bending failure in 0.1N hydrochloric acid (h)	Corrosion fatigue endurance *	
1	A	3.4	2.7	68	24	435	514	-90	>200	O	This invention
2	B	7.8	5.4	35	22	462	524	-80	>200	O	This invention
3	C	1.2	1.1	93	27	412	508	-100	>200	O	This invention
4	D	5.9	5.7	88	25	435	520	-90	>200	O	This invention
5	E	15.6	5.6	94	23	332	420	-80	>200	O	Comp. Example
6	F	8.2	5.1	77	24	562	488	-75	48	$\times$	Comp. Example
7	G	6.2	13.6	28	16	441	469	-90	>200	O	Comp. Example
8	H	5.9	5.1	68	24	426	478	-30	>200	O	Comp. Example
9	I	13.4	14.2	84	25	328	365	-60	>200	O	Comp. Example
10	J	7.2	5.4	56	24	436	467	-70	72	$\times$	Comp. Example
11	K	7.9	3.2	78	26	442	495	-75	192	$\times$	Comp. Example
12	L	7.6	1.8	85	19	415	435	-35	192	$\times$	Comp. Example
13	M	7.2	2.4	65	19	433	440	-80	>200	O	Comp. Example
14	N	7.4	5.2	49	22	345	442	-60	>200	O	Comp. Example
15	O	5.6	4.2	85	23	456	475	-75	168	$\times$	Comp. Example
16	P	6.2	6.5	55	18	468	482	-85	>200	O	Comp. Example
17	Q	3.4	2.5	62	18	436	488	-35	>200	O	Comp. Example
18	R	4.2	4.1	52	16	444	469	-90	>200	O	Comp. Example
19	S	1.5	1.6	47	17	478	478	-75	>200	O	Comp. Example
20	T	6.0	5.1	65	18	415	469	-60	>200	O	Comp. Example
21	U	5.8	6.5	52	19	433	471	-70	>200	O	Comp. Example
22	V	3.6	2.5	85	19	465	440	-35	192	$\times$	Comp. Example
23	W	4.3	5.2	25	16	551	495	-25	72	$\times$	Comp. Example
24	X	12.9	6.5	88	26	337	388	-80	>200	O	Comp. Example
25	Y	6.8	3.7	57	19	557	477	-30	48	$\times$	Comp. Example
26	Z	13.2	7.3	82	23	319	388	-85	>200	O	Comp. Example

In Table 4, in the column of corrosion fatigue endurance\*, when the fatigue life in the completely reversed plane bending fatigue test with stress amplitude  $\sigma_f$  after a salt spray test for 20 days was 1/2 or more of the number of cycles of an uncorroded specimen, the corrosion fatigue endurance was evaluated as O, and other cases were evaluated as X.

Table 5

No.	Steel	Hot rolling conditions				Tube making Width reduction of hoop (%)	Remarks
		Slab heating temperature (°C)	Finisher temperature (°C)	Slow cooling time (s)	Coiling temperature (°C)		
27	A	1230	930	6.8	660	3.2	
28	A	1140	940	2.4	690	5.6	
29	A	1190	890	2.2	700	6.5	
30	A	1210	1020	3.5	710	2.4	
31	A	1190	730	4.7	610	3.5	
32	B	1230	900	3.8	630	1.8	
33	B	1250	910	1.4	625	1.4	
34	B	1180	920	6.4	670	0.9	
35	B	1200	900	5.4	550	1.2	
36	B	1210	880	5.0	760	3.5	
37	C	1220	860	3.4	675	4.2	
38	C	1210	910	2.3	540	10.6	
39	D	1220	900	4.0	690	5.6	
40	D	1230	890	5.6	730	7.6	Hot-rolled material with scale
41	D	1190	950	2.7	630	5.2	Dimensions: $\phi 60.5 \times 2.6t$ , suspension arm
42	D	1210	880	2.2	620	2.6	Cold-rolled annealed black plate
43	D	1270	830	6.1	630	-	Press forming and welding
44	D	1200	850	3.4	690	-	Roll forming and welding
45	D	1280	900	6.0	610	3.5	Surface-treated black plate (Zn-based)
46	D	1260	940	2.5	670	5.4	Surface-treated black plate (Al-based)
47	D	1230	900	3.6	650	3.6	Zn plating after tube making
48	D	1240	960	5.9	590	-	Hot-reduced pipe material (reduction rate: 70%)
49	D	1220	870	4.6	570	-	Warm-reduced pipe material (reduction rate: 70%)
50	D	1250	830	8.5	650	5.2	Quenching during forming
51	D	1220	910	5.3	620	2.3	Shot blasting after quenching
52	D	1260	840	5.1	645	2.6	Shot peening after quenching

Table 6

No.	Steel	Ferrite grain diameter corresponding to circle, $d_f$ ( $\mu\text{m}$ )	Second phase grain diameter corresponding to circle, $d_s$ ( $\mu\text{m}$ )	Ferrite volume fraction $V_f$ (%)	Elongation of untreated tube El (%)	Cross-section hardness after quenching HV(10)	Completely reversed plane bending fatigue endurance (MPa)	Fracture appearance transition temperature in Charpy impact test vTrs ( $^{\circ}\text{C}$ )	Four-point bending failure time in 0.1N hydrochloric acid (h)	Corrosion fatigue endurance	
27	A	5.4	3.6	86	25	444	506	-90	>200	O	This invention
28	A	14.3	13.5	68	19	415	442	-70	>200	O	Comp. example
29	A	2.6	2.1	85	23	435	512	-85	>200	O	This invention
30	A	15.2	12.8	75	19	408	436	-75	>200	O	Comp. example
31	A	13.5	6.8	77	16	415	426	-65	>200	O	Comp. example
32	B	6.5	5.2	56	23	445	523	-80	>200	O	This invention
33	B	9.5	8.5	25	17	442	436	-75	>200	O	Comp. example
34	B	6.5	5.2	54	24	449	513	-80	>200	O	This invention
35	B	7.5	13.5	24	17	452	425	-70	>200	O	Comp. example
36	B	16.9	12.8	77	21	415	398	-55	>200	O	Comp. example
37	C	2.3	1.9	91	26	425	514	-95	>200	O	This invention
38	C	2.5	1.7	90	15	433	444	-80	>200	O	Comp. example
39	D	6.2	4.8	90	26	442	525	-95	>200	O	This invention
40	D	5.4	4.9	85	25	426	502	-90	>200	O	This invention
41	D	5.5	3.8	87	25	435	514	-85	>200	O	This invention
42	D	8.4	6.8	92	27	418	510	-85	>200	O	This invention
43	D	6.2	5.1	85	24	415	501	-85	>200	O	This invention
44	D	5.8	4.7	85	26	427	502	-80	>200	O	This invention
45	D	6.8	5.7	88	24	415	522	-85	>200	O	This invention
46	D	7.2	5.9	88	23	429	517	-80	>200	O	This invention
47	D	5.6	4.8	86	23	436	524	-80	>200	O	This invention
48	D	7.8	3.5	86	27	408	511	-85	>200	O	This invention
49	D	8.6	2.9	91	28	409	513	-80	>200	O	This invention
50	D	5.9	5.7	84	25	411	495	-75	>200	O	This invention
51	D	6.3	5.9	85	22	416	533	-80	>200	O	This invention
52	D	5.2	4.1	89	26	452	532	-80	>200	O	This invention